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Abstract                      Biomaterials have become an integral part of our lives as a result of the current focus on renewable

sustainability. In the realm of composites, a diverse variety of polymers coupled with multiple natural fibres enables the achievement of varied functional property criteria. Bio- composite materials have several benefits, in addition to the use of renewables, including minimal impact on the environment, light weight and biodegradability. Here we have isolated nano-silica from rice husk and fabricated woven flax fabric coated with nano-silica and phenol formaldehyde composite with varying weight percentages of nano-silica. The extracted nano-silica was confirmed from XRD, TGA and FESEM. Five types of composites (PF-F, 1 NS, 2 NS, 3 NS, 4 NS) prepared keeping constant weight of PF resin and flax fabric. Mechanical, morphological and electrical behaviour of the prepared composites were examined. From the obtained results, tensile strength was found to be maximum for 1 NS loading. In particular, as compared to neat composites (PF-F), the tensile strength of prepared specimens with nano-silica (1 NS, 2 NS, 3 NS, 4 NS) improved by 93.92%, 99.55%, 49.85% and 26.43% respectively. The FESEM pictures of fracture surfaces demonstrated that the inclusion of nano-silica boosted fibre's interfacial strength, strengthened both fibre and matrix, and improved resin adhesion to fibre, therefore enhancing the composite tensile characteristics. Due to polarisation processes mediated by nano-silica inclusion, the dielectric constant in PF hybrid composites increases as the loading of nano-silica increases. Electronic polarisation causes an increase in AC conductivity at high frequencies (3 NS). The graph shows that as the amount of nano-silica loaded rises, the AC conductivity increases. The dielectric constant values in 2 NS are the highest.

*Graphical Abstract:*



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Keywords (separated by '-') Nano-silica - Phenol-formaldehyde resin - Flax fabric - Hybrid composite - Mechanical - Fracture morphology - Electrical properties

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Footnote Information

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# 2 Mechanical, Morphological Behaviour and Electrical Conductivity 3 of Phenol Formaldehyde-Flax Fabric (PF-F) Hybrid Composites 4 Reinforced with Rice Husk Derived Nano-silica

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## AQ1 AQ2 Abstract

10 Biomaterials have become an integral part of our lives as a result of the current focus on renewable sustainability. In the  
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12 functional property criteria. Bio- composite materials have several benefits, in addition to the use of renewables, including  
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26 **Keywords** Nano-silica · Phenol–formaldehyde resin · Flax fabric · Hybrid composite · Mechanical · Fracture morphology ·  
27 Electrical properties

## 1 Introduction

29 Polymers have a numerous benefit which allows polymeric materials and composite materials, to make friction parts  
30 like bearings, wheels, piston rings, and soft seals. These are easy to fabricate, ultra-lightweight and resistant to  
31 chemical and environmental effects. Fabric reinforced polymer composites, as compared to other polymer composites,  
32 provide increased mechanical properties in both the horizontal and vertical directions of the fabric, as well  
33 as the ability to adhere to curve surfaces without tangling [1, 2] Fabric reinforced polymer composites, as a result  
34 of their large use in the fields of aircraft, aviation, high-speed railways, automotive, and other fields, have recently  
35 sparked a broad range of financial and research interest.

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The polymer used is the commanding factor which plays the key role in determining the properties of the hybrid composite. If fire, smoke, and toxicity (FST) conditions must be taken into consideration, phenolic resin have a considerable market share compared to epoxy resin. They're also employed as ablative composites for thermal protection in aerospace applications like solid rocket motor nozzles and heat shields for space vehicle atmospheric re-entry. There are several prepreg conditions that may be discovered here. Because of the phenolic resin's chemistry, phenolics have a substantially longer storage life than most epoxy prepreps. This is due to the nature of the phenolic resin's chemistry [3, 4].

Natural fibres have recently received much attention as a replacement for synthetic fibres in FRP (fibre reinforced polymer) composites, attributed to high environmental issues and a high demand for eco-friendly materials [5]. Flax is one of the natural fibres that has desirable material properties and could be used to replace glass fibres in fibre reinforced polymer (FRP) composites. Flax fibre provides the best possible combination of low cost, light weight, and high tensile strength for structural applications [6–8]. According to Yan et al. [9] flax, hemp, and jute are the three most excellent materials for substituting glass fibres in terms of cost, mechanical efficiency, and production yielding. Flax fabric/epoxy composite has a tensile strength of 300 MPa, which is comparable to GFRP composite, according to Assarar et al. [10] automotive engineering is currently a wide demand market for natural FRP composites. The advantage associated with natural FRP composites are it reduce the energy required for production by 80% and minimises component mass [11]. Natural FRP composites have strength and durability, therefore it can be used as crashworthy structures for the manufacture of vehicles.

From the perspective of energy absorption, Yan et al. [12] recently investigated the crash safety of flax fabric reinforced epoxy composites. Flax fabric/epoxy composite were discovered to have the capability to be applied as energy absorber devices. Comparing with traditional metallic or G/CFRP composite tubes flax fabric/ epoxy composite tubes with foam-filler absorbed more energy during axial and lateral crushing [13, 14]. After comparing the energy absorption capacities of composites reinforced by unwoven hemp, woven flax, and jute, Meredith et al. [15] suggested flax/epoxy composites to be used as energy absorbers for vehicle construction. The most important application of natural FRP composites is probably as construction materials. These renewable products would help to lower the costs and improve energy consumption, addressing urgent infrastructure needs while also encouraging sustainability. Natural fibres/fabric composites may be used in construction as a hybrid framework in combination with traditional construction materials, such as a flax fibre reinforced polymer

(FFRP) tube encapsulated coir fibre reinforced concrete (CFRC) structure (FFRP-CFRC) [16].

Natural fibres, known as cellulosic biomass, such as banana, curauá, coconut, flax, hemp, jute, pineapple, ramie, sisal, and sugar cane bagasse, are often used as filler in composite material since they provide numerous advantages over synthetic fibres, including: relatively low mass; easy handling; protection thermal, electric, and acoustic; architectural features; non-toxic; and low specific mass [17–19]. Within the automotive sector, it is thought to have a bright future being used in the interior coating of vehicles, buses, and trucks; in construction works to reinforce cement; as a woven fabric; in aerospace, sport, and packaging, indicating a booming market [20, 21]. For applications in engineering sector the durability issues should be rectified primarily. In practical use, these bio-composites would be subjected to a variety of extreme conditions, including atmospheric and weather conditions, resulting in composite material deterioration and improving safety issues. As a result, learning about the stability of flax FRP composites may be useful for applying in practical uses [22]. But nevertheless, technological problems prevent widespread use of these composites [23].

The most significant are fibre cultivation inconsistencies, composite manufacturing inconsistencies, and a lack of clear information of their behaviour. Due to a variety of factors such as temperature, water, radiation, fungus/bacteria, and mechanical stress using such products in the indoor and outdoor applications causes particular ageing issues leading to total deterioration. Joseph et al. [24] studied the mechanical properties of sisal fibre-polymer composites subjected to water sorption and UV degradation to investigate the performance of natural FRP composites. Many researchers investigated the properties of natural fibre reinforced polymer composites [25–27].

An increasing tendency in the fabrication of hybrid nanocomposites, which has sparked widespread interest and awareness nowadays. Nano clay layers, nanotubes, nano-silica and spherical particles (metal nanoparticles) are examples of inorganic materials used in nanocomposites. The uniqueness of nanomaterials are they have the potential to change the polymer's properties by altering the mobility of the polymer chains. One of the promising nanomaterial is inorganic silica nanoparticles which can enhance the thermo-mechanical properties of a polymer [28, 29]. The peculiar properties of nano-silica are due to the surface chemistry characterised by the presence of silanol groups in silica nanoparticles.

Dinesh et al. [30] investigated mechanical, thermal and morphological behaviour of nano-silica treated pineapple woven fabric in epoxy resin. From the mechanical tests they found that 1 vol% nano-silica and 30 vol% pineapple fibre gave maximum properties. The thermal degradation was slowed down by the addition of 1 vol% nano-silica. Same was the trend

for tribological behaviour. The fracture toughness was also improved by the nano-silica addition which was confirmed from the SEM images. Despite the fact that silica has many applications in polymers, the agglomeration and low compatibility of SiO<sub>2</sub> in the rubber matrix leads to its restricted applications. The surface treatment of sisal fibres improved the interface adhesion between fibres and resin by adding nano-silica to the phenol formaldehyde resin [31]. Under various temperatures, the borax-treated fibres demonstrated the highest heat resistance, constant friction coefficient, and low wear rate. The best fibre content was 15 wt.%. The impact of nano-silica and nano-clay on glass fibre reinforced composite have been studied [32]. It is for this reason that adding nano-silica and/or nano-clay to GFRP composites may help them perform better in corrosive conditions.

As a result of their higher biocompatibility, reusability, and degradability compared to synthetic fibres, natural fibres (such as coconut fibres, silk, flax fibres, and flax seed fibres) and synthetic fibres (such as glass fibres) are often used to reinforce polymer or hybrid or natural resin matrix composites [33]. Epoxy resin is blended with *Lannea coromandelica* (LC, Anacardiaceae plant gum) to create hybrid *Lannea Coromandelica* Blender Epoxy matrix composites (LCE) that are biodegradable and environmentally friendly. Hybrid Composites (Pineapple/Silk/Flax fibre mats with 2%, 4%, and 6% volume fraction of Bentonite nanoclay in each composition reinforcement in hybrid LCE resins prepared by compression hand moulding process) and hybrid LCE resin are made using this approach. Hybrid composites' mechanical properties have improved by three to four times when compared to hybrid LCE resin. DMA findings show that untreated and treated hybrid composites have the highest storage modulus and lowest damping factor compared to hybrid LCE resin. LCE resin composites reinforced by P/G/P fibre mats can be biodegraded at a 4% volume fraction of nano filler, according to biodegradability studies. In light of the increasing environmental, economic, and application issues, hybrid composites are expected to garner substantial attention in the future [34, 35]. As an alternative to synthetic fibres, researchers developed novel composites that included more than one reinforcement from natural resources, such as natural fiber/natural fibre or natural fiber/nanofiller from organic sources [36]. It is important to use natural fibres and fillers instead of synthetic ones in order to reduce the environmental impact of our products. There has been a rise in the use of natural filler/fibers as a reinforcing mediator because of the abundance of these materials [37].

So far, there have been very few studies on the property studies (mechanical, morphology and electrical properties) of flax fabric polymer composites. According to a survey of the literature, no particular research has been done to evaluate the influence of a PF matrix containing nano-silica (extracted from rice husk) on the mechanical, morphological and electrical behaviour of Flax fabric/PF hybrid composites at the same time. Natural fibres and natural fillers that are environmentally

friendly are becoming more popular, and they should be used instead of synthetic fibres and fillers to meet the growing demand for green products. Because natural filler/fibers are readily available, the use of natural filler/fiber as a reinforcing mediator has evolved in the application of science and technology. The combined effect of nano-silica on the mechanical, morphological behaviour and electrical properties of flax fabric-phenol formaldehyde hybrid composites are examined in this study. The prepared hybrid composites have the potential for large-scale applications because to their commercial availability, inexpensive raw material, and simple production procedure. The method we have developed is an easy, low-cost, and ecologically friendly method. This paper describes the fabrication of an environmentally friendly PF-flax fabric hybrid composites using compression moulding. Nano-silica extracted from rice husk were utilised to increase the hydrophobicity of the materials and to improve their dimensional stability. When the nano-silica is added, the properties of the products were greatly enhanced. This research was able to increase the mechanical, and electrical properties of the PF hybrid composite, and it has implications for the industrial manufacturing as well as applications in high-humid environments.

## 2 Materials and Methods

### 2.1 Materials

Rice husk was obtained from local sources (rice mill, Kalady, Kerala, India). Rice husk contains impurities such as alkali metals, and organic constituents. Sodium Hydroxide, Hydrochloric acid, Ethanol of analytical grades (AR) were purchased from Merk, India. Plain woven flax fabric was obtained from W.F.B. Baird & Company, Kochi, Kerala, India (Table 1). Phenol formaldehyde resin (PF) was purchased from Polyformalin, Ernakulam, Kerala, India. The pH of PF is 9–10, specific gravity at 30 °C is 1.252 and total solid content is 49.68% w/w.

### 2.2 Extraction of Nano-silica from Rice Husk

The raw material rice husk was obtained from rice mill at Kalady, Ernakulam, Kerala. All reagents used were of analytical grade, and their solutions were made up in distilled

**Table 1** Basic parameters of flax fabrics

Parameters	Flax fabric
Woven style	Plain
Density (gm/cc)	1.3
Thickness (mm)	0.45
Weight (gsm)	140

239 deionized water. The rice husk was thoroughly cleansed so  
 240 as to eradicate sandy and dust particles. The extraction pro-  
 241 cess was adopted from published literature, with a few minor  
 242 alterations, to meet our needs [38, 39]. Later it was subjected  
 243 to heat treatment to obtain the ash. Samples were burned  
 244 inside a programmable muffle furnace at 700 °C for 2 h. Fol-  
 245 lowed by the addition of 1.5 M sodium hydroxide solution.  
 246 Then the filtrate was treated with conc. HCl to form the silica  
 247 sol. The suspension was filtered and washed several times and  
 248 kept in oven for 24 h at 70 °C. The dried powder was calcined  
 249 at 700 °C for 2 h and used for further studies.

### 250 **2.3 Preparation of Nano-silica Coated Flax Fabric/ 251 Phenol–Formaldehyde Composites**

252 PF-flax fabric hybrid composites were fabricated by the com-  
 253 pression moulding technique. The weight of nano-silica for the  
 254 preparation of PF hybrid composites was taken based on the  
 255 total weight of PF and flax fabric (Table 2.). Ethanol is used  
 256 as a dispersing agent for nanomaterials. To obtain a uniform  
 257 dispersion, the solution was homogenised for half an hour. Flax  
 258 fabric was cut into a square piece (15\*15 cm). The nanomater-  
 259 ial dispersion was sprayed on the fabric and kept in the oven  
 260 at 70 °C for the removal of ethanol. On the dried fabric, previ-  
 261 ously weighed PF resin was poured and spread using a roller.  
 262 After that, the sample is pressed in a pre-heated compression  
 263 moulding machine at 100 °C for 30 min. The sheets of hybrid  
 264 composites were produced and cut into the desired size after  
 265 cooling, and were used for further research.

### 266 **2.4 Characterisation of Nano-silica Coated Flax 267 Fabric/Phenol–Formaldehyde Composites**

#### 268 **2.4.1 X-ray Diffraction Analysis (XRD)**

269 The crystallinity of the nano-silica and PF hybrid compos-  
 270 ites were recorded on Bruker AXS D8 Advance with Cu  
 271 K $\alpha$  radiation with an angle range 5°–80° (2 $\theta$  angle range)  
 272 at a wavelength of 1.541 Å and an operating voltage of  
 273 45 kV and a current of 35 mA.

**Table 2** Designation and formulation details of prepared PF-F hybrid composites

Sample code	PF resin (g)	Flax fabric (g)	Nanofiller loading (g)	Weight percentage of composites
PF-F	10	10	0.0	0.00
1 NS	10	10	0.1	0.02
2 NS	10	10	0.2	0.04
3 NS	10	10	0.3	0.06
4 NS	10	10	0.4	0.08

#### 274 **2.4.2 Field Emission Scanning Electron Microscopy (FESEM)**

275 The morphology of nano-silica and fracture surface of PF  
 276 hybrid composites were analysed using Hitachi SU6600  
 277 Variable Pressure Field Emission Scanning Electron  
 278 Microscope (FESEM) at acceleration voltage of 30 kV  
 279 and Probe current of 1pA ~ 200nA. All the samples were  
 280 sputter-coated with gold to avoid charging.

#### 281 **2.4.3 Thermogravimetric Analysis**

282 The thermal properties of the samples were analysed using  
 283 Perkin Elmer Diamond TG/DSC. In a nitrogen atmos-  
 284 phere, 10 mg of the sample was placed on the alumina  
 285 cup and heated at a rate of 20 °C /min. The temperature  
 286 range of measurement is from 25 °C to 700 °C.

#### 287 **2.4.4 Atomic Force Microscopy (AFM)**

288 The surface morphology of pristine nano materials and  
 289 composites were characterised with Atomic Force Micros-  
 290 copy (WITec GmbH, Ulm, Germany) in contact mode at  
 291 room temperature. The suspension of CNT and CNT-  
 292 COOH was sonicated well and then analysed.

#### 293 **2.4.5 Mechanical Characterisation**

294 The mechanical properties of the samples were tested  
 295 using Universal Testing Machine (Tinius Olsen) accord-  
 296 ing to ASTM D 638. The samples were cut into rectangular  
 297 strips and the testing were conducted at room temperature  
 298 with a gauge length of 60 mm and speed rate 2 mm/min.

#### 299 **2.4.6 Electrical Conductivity**

300 The dielectric properties of the samples were measured  
 301 using Wayne Kerr 600B precision LCR meter with fre-  
 302 quency ranging from 20 Hz to 30 MHz.

## 303 **3 Results and Discussion**

### 304 **3.1 X-ray Diffraction (XRD) of Nano-silica**

305 The XRD pattern of nano-silica isolated from rice husk  
 306 (Fig. 1) showed a wide-angle at 2 $\theta$  = 22°, confirming nano-  
 307 silica formation from rice husk. This unique broad peak at  
 308 2 $\theta$  = 22° indicated the presence of silica in the amorphous  
 309 form [40–42]. The obtained broad diffraction peak clearly



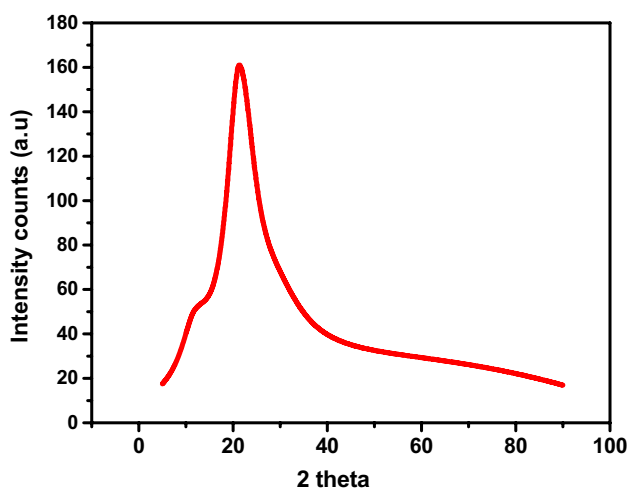
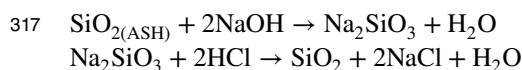


Fig. 1 XRD curve of nano-silica from rice husk

310 shows the existence amorphous nano-silica. If the peak was  
 311 sharp then the silica obtained should be crystalline. The pure  
 312 nano-silica, on the other hand, was sintered at 1373 K to  
 313 reach the crystalline phase and quantify the crystallite size.

314 By treating with sodium hydroxide, sodium silicate is  
 315 formed. When concentrated HCl is added pure nano-silica  
 316 is formed.



319 The presence of only one peak confirms the removal  
 320 organic constituents and alkali metals associated with silica  
 321 [43]. Chidambaram et al. [44] obtained similar diffraction  
 322 peaks for nano-silica derived from groundnut shell, coconut

323 husk, banana peel, walnut shell and orange peel. Previous  
 324 findings demonstrated by other researchers, in which bio-  
 325 genic nano-silica was produced using rice husk as a precur-  
 326 sor [45, 46]. The XRD peak of the produced biogenic silica  
 327 revealed that it was amorphous in form, which is closely  
 328 related to the current study.

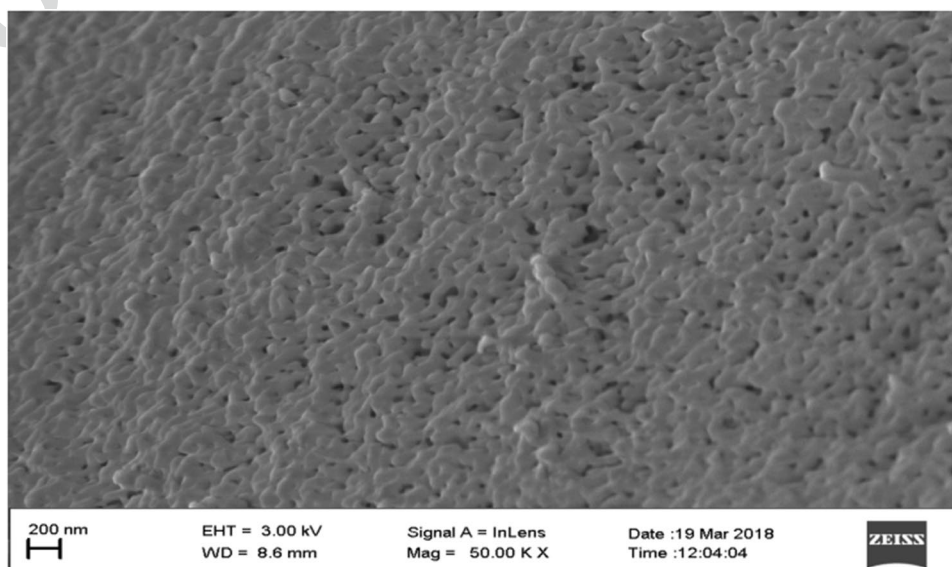
### 3.2 Field Emission Scanning Electron Microscopy (FESEM): Nano-silica

329 From the FESEM analysis (Fig. 2) it was observed that  
 330 rice husk-derived nano-silica exhibited agglomerations.  
 331 The hydrogen bonding between silanol groups on the  
 332 surface of nano-silica was determined to be the cause of  
 333 aggregation. The particles were found to be spherical and  
 334 aggregation of silica-silica was found to be minimal. This  
 335 result is in agreement with other literature reports [47]. Jen  
 336 et al. [48] have isolated silica nanoparticles from palm ker-  
 337 nel shell ash and analysed the morphology. SEM images  
 338 showed porous nature of silica nanoparticles similar to  
 339 our result. These bio-derived nanoclusters can be used as  
 340 carriers in biomedical and nutraceutical applications [44].  
 341  
 342

### 3.3 Thermogravimetric Analysis (TGA): Nano-silica

343 Here we can see that weight loss was little with an increase  
 344 in temperature (Fig. 3). Initial weight loss was due to the  
 345 removal of residual moisture [42, 49, 50]. The very slow  
 346 weight loss can be attributed to the porous nature of nano-  
 347 silica. As a result, nano-silica required more time to heat  
 348 up before water molecules could be released from the  
 349 nano-silica.  
 350

Fig. 2 FESEM image of nano-silica obtained from rice husk



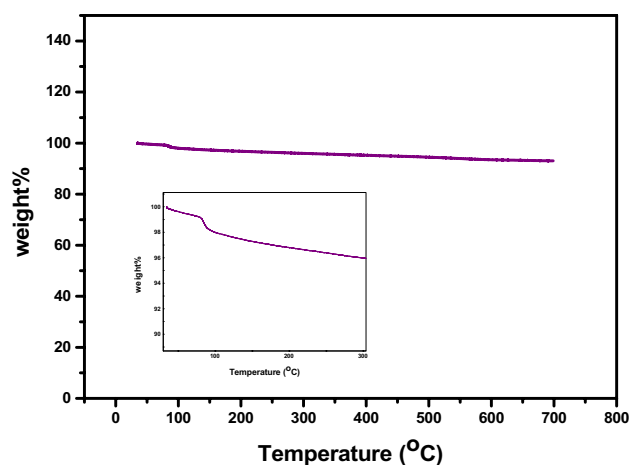


Fig. 3 TGA of nano-silica

### 3.4 X-ray Diffractogram (XRD) Analysis

The XRD data is used to determine the structural properties of PF hybrid composites with nano-silica. The XRD of PF hybrid composites with different loadings of nano-silica is shown in Fig. 4. The major peaks are observed at  $18^\circ$  and  $22^\circ$ . Pure PF does not show any strong peaks. The peak of nano-silica is at  $22^\circ$ . All PF hybrid composites have identical XRD patterns. The XRD results further show that the interfacial interaction of nano-silica with the PF-F matrix caused distinctive variations in crystallinity. The interfacial interaction of nano-silica with the PF-F matrix produced a characteristic improvement in the crystallinity of the hybrid composites.

### 3.5 Atomic Force Microscopy (AFM)

The surface topographies of PF-F and 2 NS is shown in Figs. 4 and 5. Atomic force microscopy is used to look at

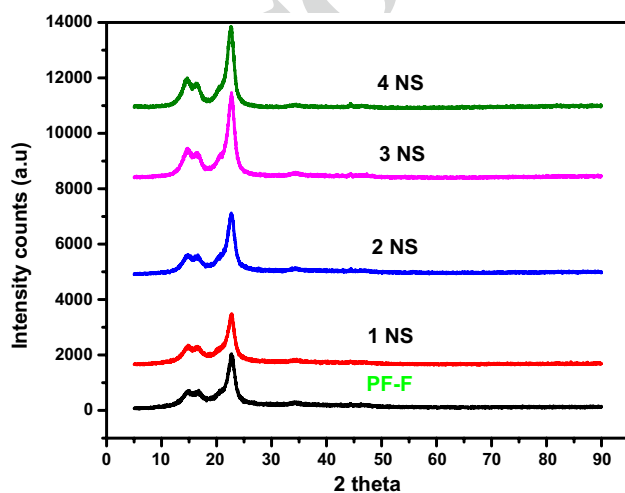


Fig. 4 XRD spectrum of PF-F hybrid composite with nano-silica

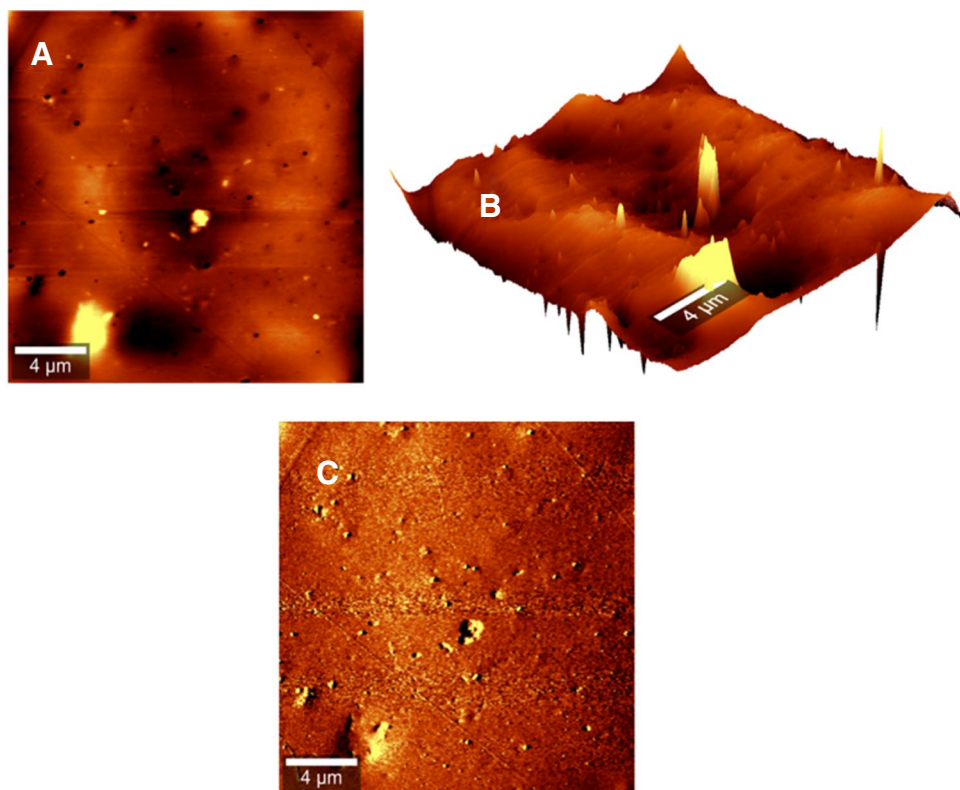
the surface roughness of composites that have been made. Figure 5A-C shows the AFM images of PF-F composites. This shows slightly rough surface. Figure 6A-C shows the AFM images of 2 NS composites. This loading was selected because it provides superior features in terms of sorption properties, hydrophobicity, thermal stability, and other characteristics. According to the enhanced properties of PF-F hybrid composites, as demonstrated in the experiment, the figure shows that the nano-silica were spherical, with minor agglomerations. One of the most important factors contributing to the better performance of produced composites is the fine dispersion of nanomaterials in the matrix. It has been demonstrated that the roughness of the hybrid composites prepared decreases as a result of the addition of nano-silica from 8.58 nm (PF-F) to 7.31 nm (2 NS). It is possible that the enhanced roughness of the surface of PF-F's is responsible for the lower contact angle and better wettability. Hence decreased roughness will lead to less wettability [51].

### 3.6 Mechanical Properties of PF-F Hybrid Composites Reinforced With Nano-silica

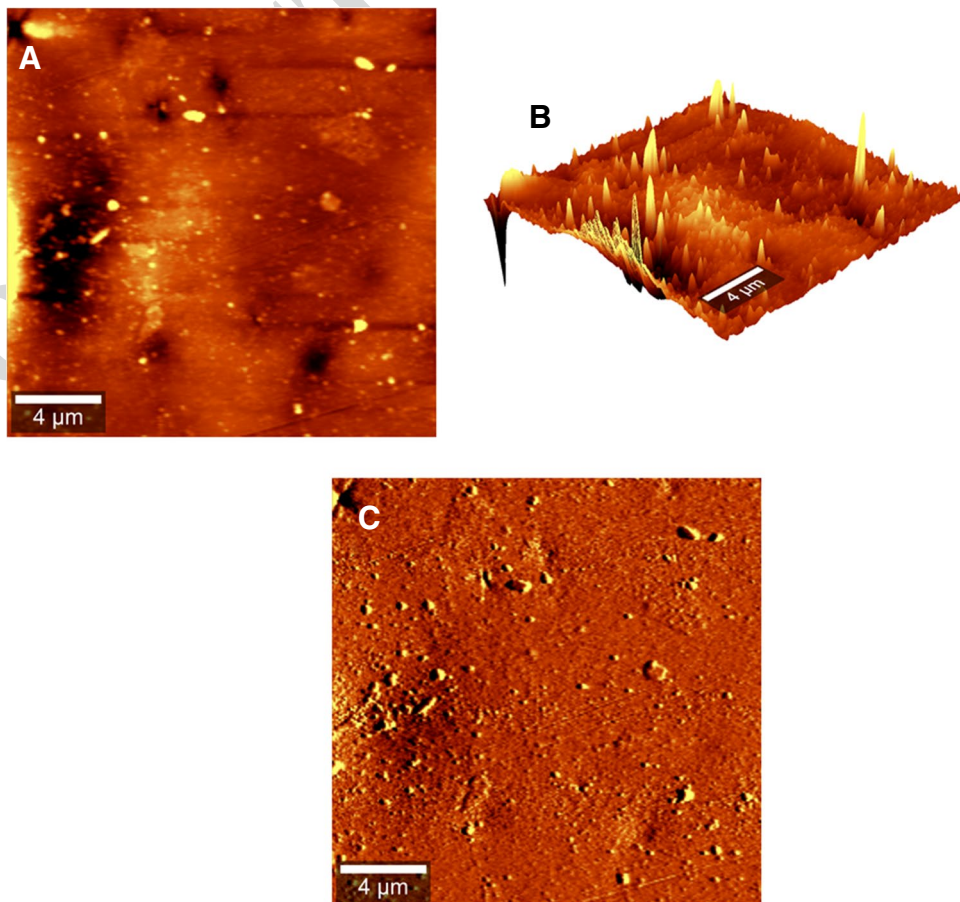
The stress-strain behaviour of woven flax fabric-PF composites and nano-silica coated composites is shown in Table 3. From the stress-strain curve, it is observed that the stress increases steadily with strain until all specimens break. It is obvious from the table that, in the case of brittle material, the stress-strain curve approaches linearity almost all the way to failure. At low strains, the stress-strain curves of the composites exhibit linear behaviour; nevertheless, when the composites fail completely, a considerable shift in slope is detected, indicating that the composites are exhibiting nonlinear behaviour. The stress-strain behaviour becomes more pronounced with the addition of nano-silica (2 NS). Due to the excellent load transmission between the nano-silica and the PF matrix and flax fabric, the composite 2 NS exhibits the highest tensile stress value. With the addition of nano-silica, the stiffness of the PF hybrid composites improved. The composites became more brittle by the addition of nano-silica, and before the composite ruptured, the elasticity of the matrix had been significantly enhanced. Here, the maximum energy needed for rupturing is  $3\text{ NS} > 4\text{ NS} > 1\text{ NS} > 2\text{ NS} > \text{PF-F}$ .

The tensile strength of the PF hybrid composite was the least, while nano-silica inclusion increased the composite mechanical properties. Because 2 NS has good tensile strength, it will experience less deformation and degradation during testing and can withstand higher stress, which will improve its tensile strength. It should be emphasised that all samples showed brittle failure characteristics. The nano-silica content plays a crucial role in improving mechanical properties. The addition of fibres, reinforcements, and additives in the matrix is known to improve

**Fig. 5** **A-C** AFM image of PF-F composite **B** 3D image of PF-F composite



**Fig. 6** **A-C** AFM image of 2 NS **B** 3D image of 2 NS



**Table 3** Mechanical properties of PF-F hybrid composites with nano-silica

Sample code	Tensile strength (MPa)	Young's modulus (MPa)	Elongation at break (%)
PF-F	31 ± 1	1010 ± 2	8 ± 1
1 NS	61 ± 2	1095 ± 7	11 ± 1
2 NS	63 ± 2	1217 ± 4	10 ± 1
3 NS	47 ± 1	1051 ± 4	22 ± 1
4 NS	39 ± 1	1025 ± 6	20 ± 1

417 composite mechanical properties. By increasing nano-  
418 silica concentration, the tensile strength of composites  
419 improves. The tensile strength of the flax/PF composite  
420 is initially 31.59 MPa, but the addition of 1 NS raises the  
421 tensile strength to 61.26 MPa. That is, it shows an increase  
422 of 93.92%. The tensile strength is at its maximum observed  
423 for 2 NS, which is 99.55% higher than the PF-F com-  
424 posite. The tensile strength of 3 NS and 4 NS is 47.34 MPa  
425 (49.85%) and 39.94 MPa (26.43%), respectively.

426 The even distribution of nano silica enhances the  
427 weight carrying capabilities of composites. The nano-  
428 silica in nano-silica-coated flax/PF composites works as  
429 a deformation barrier, increasing composite strength. The  
430 tensile strength of composites is increased as a result of  
431 the formation of a strong interface between PF, nano-sil-  
432 ica, and flax fabric [52–54]. At higher loading, the reduc-  
433 tion in tensile strength was observed due to agglomera-  
434 tion, resulting in ineffective stress transfer. A tremendous  
435 improvement in tensile strength was observed due to the  
436 nano silica addition since an effective stress transfer is  
437 possible from nano-silica coated flax fabric to the PF  
438 matrix. The primary elements of flax fibre are cellu-  
439 lose, hemicellulose, wax, lignin, and pectin, in different  
440 ratios. The mechanical properties of the composite are  
441 determined by the cellulose, the toughest and strongest  
442 chemical ingredient in the fibre [9]. In all natural fibres,  
443 the cellulose is wrapped in non-cellulosic ingredients.  
444 The inclusion of silica nanoparticles into the structure of  
445 the fibre improves the fibre's load bearing capability of  
446 PF-Flax fabric composites. At higher loadings of 3 NS  
447 and 4 NS, the tensile strength decreased gradually due to  
448 the ineffective load transfer. At higher loading there is a  
449 chance of agglomeration, which results in the formation  
450 of localised stress points. This results in a lowering of  
451 mechanical properties.

452 The higher amount of nano-silica in the composite  
453 could operate as a stress concentration zone, resulting in  
454 localised cracks that inhibit the composite from improv-  
455 ing its properties. A better particle dispersion order effi-  
456 ciently restrains the load flow [55]. The aggregation of  
457 particles causes poor dispersion quality at higher loading

of nano-silica. The tensile strength reduces as a result  
[56–59]. Because of the improved fibre-matrix interac-  
tion caused by good adhesion, the tensile strength and  
modulus of a fibrous composite system were boosted.  
The reduction in free volume contributes to the improved  
tensile characteristics of prepared composites. The nano-  
silica loading improved the elastic nature PF-flax fabric  
composites.

The elastic behaviour of the composites is determined  
from the percentage of elongation. The highest elongation  
at break was observed for 3 NS and at higher loading of  
nano-silica (4 NS) the elongation at break slightly decreased  
than 3NS. The 1 NS and 2 NS composite have toughening  
behaviour compared to other loadings which is evident from  
less elongation at break. The addition of nano-silica at lower  
loading improves the stiffness of the entire composite thereby  
reducing less plastic deformation. From the results we can  
conclude nano-silica addition improved mechanical prop-  
erties. The tendency for elongation at break (%) to decline  
and eventually break at higher filler loading, matrix defor-  
mation is not only influenced by the nature of the interface  
but also by the dispersion of the fillers. Due to the addition  
of mechanical constraints by the nearly indeformable nano-  
silica particles, the modest fall in ductility implies a reduction  
in matrix deformation. Nano-silica act as stress concentrators  
due to various elastic characteristics of the material's con-  
stituent parts. Particle agglomeration encourages increased  
cavitation, higher stress concentration, and faster breaking  
[60]. Many researchers got similar trend with the addition  
of nano-silica in other polymer matrices [61, 62]. Feli et al.  
[63] got improvement in mechanical properties with nano-  
silica loading in epoxy composites. Zhou et al. [64] found  
enhancement in elongation with the addition of 2 wt.% nano-  
silica in epoxy-carbon fibre composites. In correlation with  
stress–strain curve the toughness was calculated and found  
that nano-silica addition improved the same. The mechanism  
of interaction is schematically shown in Fig. 7.

### 3.7 Fracture Mechanism by Morphology Analysis (FESEM)

The fracture surface morphology of PF-flax fabric compos-  
ites and nano-silica coated flax fabric/PF composites were  
evaluated under a scanning electron microscope (Fig. 8).  
When a polymer is used, it transmits the load to the fabric  
and functions as structural support for the entire composite.  
As a result of the loading, the composite is pulled and the  
composite's elongation occurs. Elongation causes cracks to  
appear in the material, and the brittleness of the PF will tend  
to spread the locally formed crack while the material is being  
stretched. As a result, a piece of the polymer surrounding the  
fibres is removed throughout the procedure. This is respon-  
sible for the debonding of the fibres. As the fibre de-bonds,

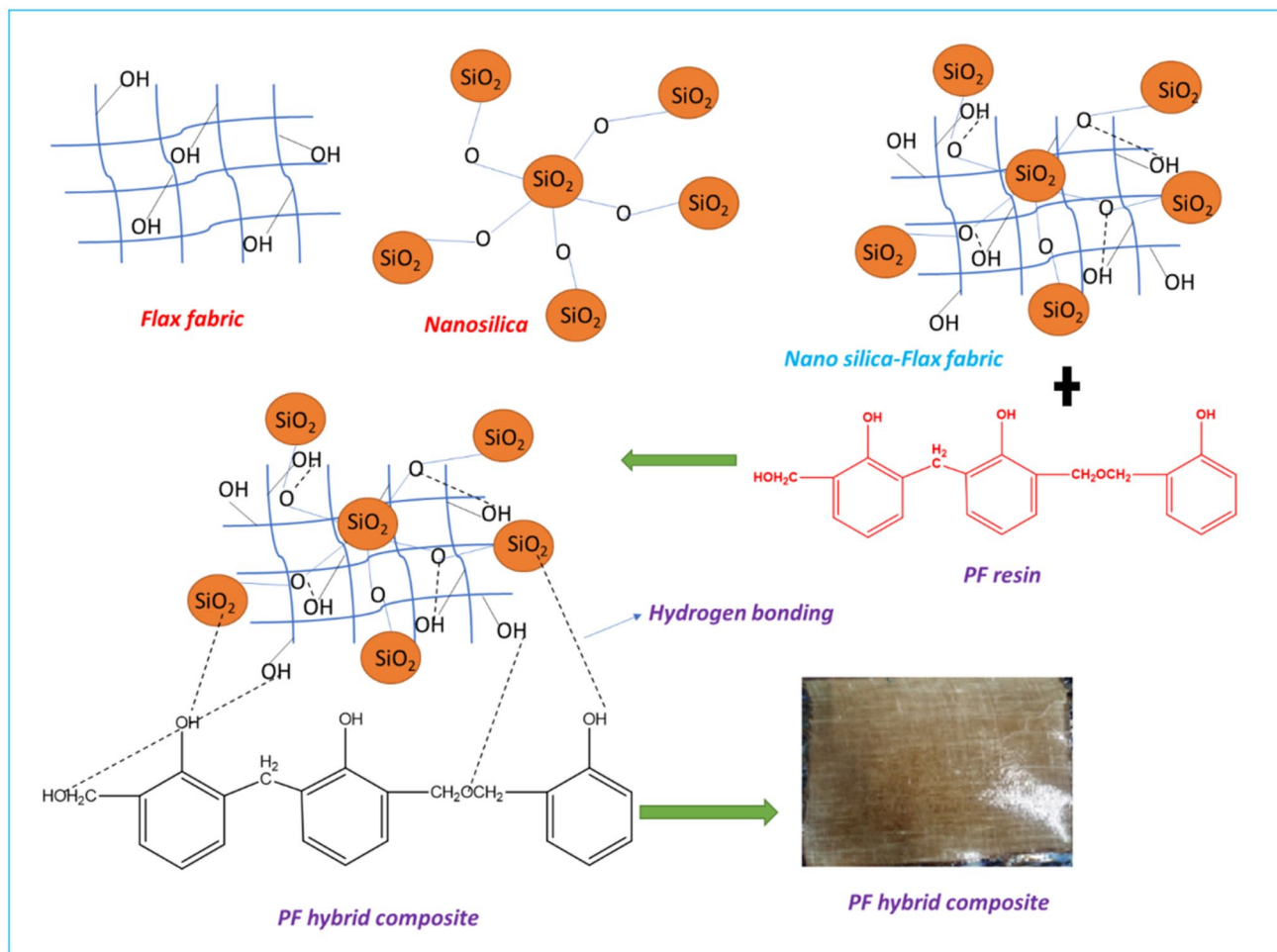


Fig. 7 Plausible mechanism of interactions present in PF-F hybrid composite

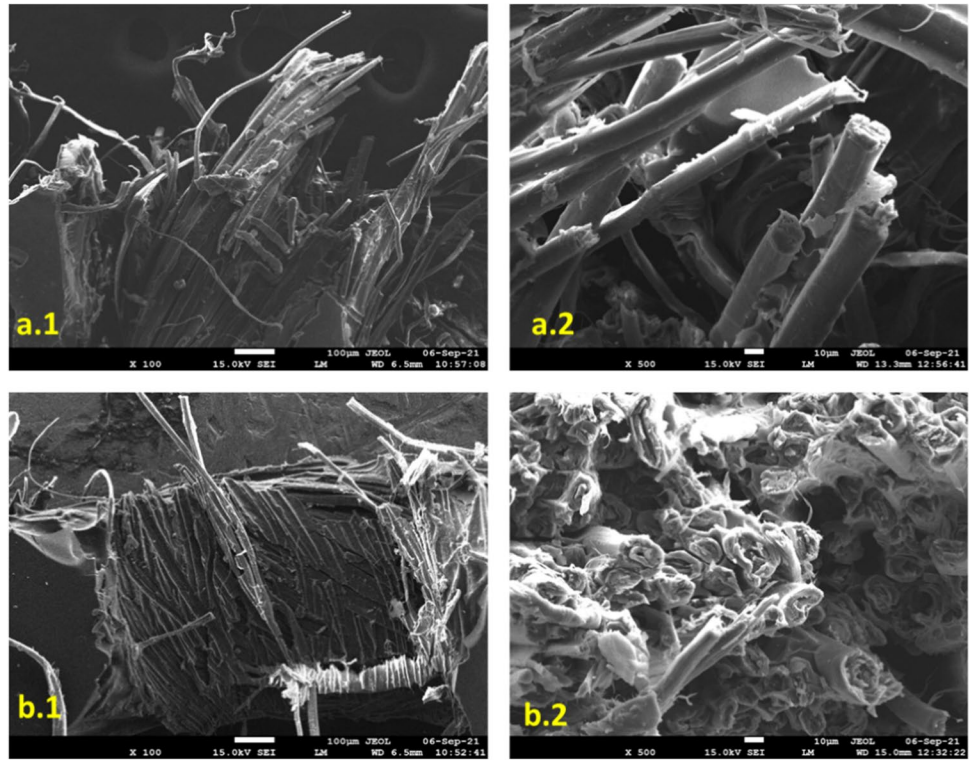
509 it progressively loses its capacity to bear weight, and, as a  
510 result, the material's strength diminishes with time.

511 Figure 8a.1 & a.2 shows fibre pulling out from the fab-  
512 ric and debonding from the PF-F matrix. The presence of  
513 nano-silica can limit the growth of microcracks caused by  
514 stress concentration by having good interaction and a strong  
515 bond with the PF matrix and flax fabric (Fig. 8b.1 & b.2).  
516 Therefore, the toughness of hybrid composites is improved.  
517 It's critical to have effective stress transmission between the  
518 PF, flax fabric, and nano-silica in order to obtain maximum  
519 mechanical properties. The existence of interfacial interac-  
520 tion enhances the passage of stress within the composite  
521 [65]. Composites reinforced with nano-silica reinforcement,  
522 on the other hand, show evidence of better fabric-matrix  
523 bonding [66]. On closer inspection, it can be observed that  
524 the nano-silica reinforcement ensures better load-bearing  
525 efficiency. There are no apparent fibre pull-outs, indicating  
526 that the de-bonding of fibres has been greatly decreased. As  
527 a result, increasing the load-bearing capability of nano-silica  
528 can be advantageous.

### 3.8 Electrical Properties of PF-F Hybrid Composites Reinforced With Nano-silica

529 The polarizability of a substance has an effect on the dielectric  
530 constant of that material. The dielectric constant of a polymeric  
531 material is affected by polarisation at the interface, at the dipole,  
532 at the electronic level, and at the atomic level [67]. Due to the  
533 presence of polar groups in PF resin, it has a dielectric constant.  
534 The interfacial polarisation is also present in hybrid compos-  
535 ites since they are heterogeneous. Because the molecule may  
536 be fully oriented at low frequencies, the dielectric constant has  
537 a frequency dependency. Medium-frequency transmissions  
538 provide little time for orienting. At extremely high frequencies,  
539 molecular orientation is impossible. Here we can see that due to  
540 the presence of nano-silica, the dielectric properties have been  
541 improved. This is because of the polar groups (nano-silica) pre-  
542 sent in the hybrid composite. 2 NS has highest AC conductiv-  
543 ity [Fig. 9a, dielectric conductivity Fig. 9b and dielectric loss  
544 Fig. 9c]. This was most likely due to a higher population of  
545 nano-silica in the backbone, as well as phase separation. The  
546  
547

**Fig. 8** FESEM images of **a.1** & **a.2** PF-F and (lower and higher magnification) **b.1** & **b.2** nano-silica coated PF-F hybrid composites (2 NS) (lower and higher magnification)



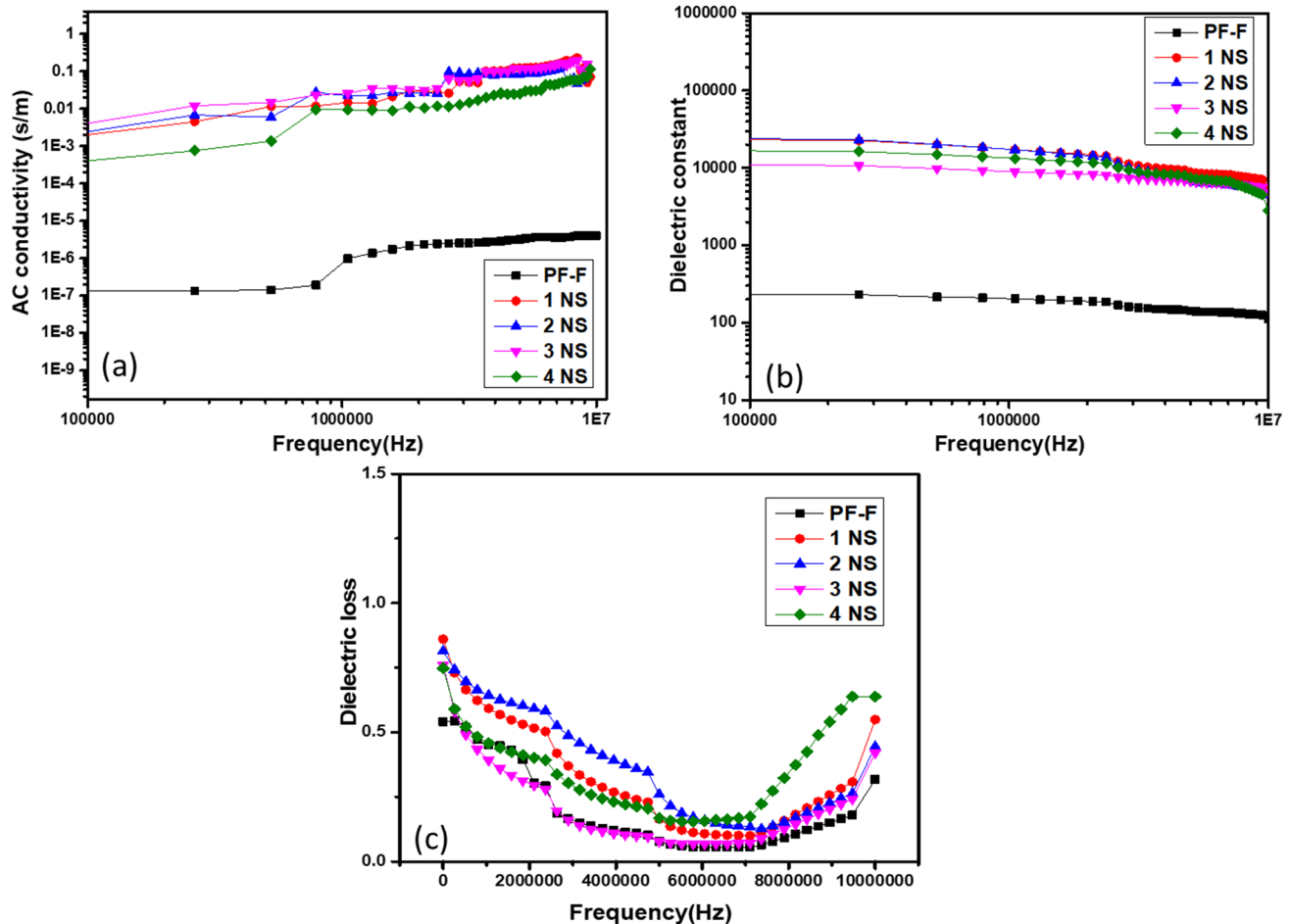
548 total dielectric constant of the PF hybrid composite was greater  
 549 than the PF-F composite because nano-silica has a higher dielec-  
 550 tric constant than pristine composite [68]. It is apparent that  
 551 when the frequency of application is increased, the ac conduc-  
 552 tivity rises. This might be attributed to an increase in absorbed  
 553 energy, which in turn leads to an increase in the number of  
 554 charge carriers involved in the conduction process. These find-  
 555 ings corroborate previous findings for various polymer compos-  
 556 ites [69–71].

557 Interfacial, orientation, atomic, and electronic polariza-  
 558 tions all have a role in determining a composite's dielec-  
 559 tric constant (E). The matrix and filler conductivities, or  
 560 polarizations, vary, resulting in interfacial polarisation in  
 561 composites [72]. When polymers with polar groups are put  
 562 in an electric field, orientation polarisation occurs. Filler  
 563 concentration affects both the composite's orientation and  
 564 interfacial polarisation. At low frequency, nano-silica load-  
 565 ing improved the dielectric constant as seen in Fig. 9b. Natu-  
 566 ral fibres include polar groups of cellulose, which contrib-  
 567 ute to an increase in orientation and interfacial polarisation  
 568 with increasing fibre loading. The E exhibits greater values  
 569 at lower frequencies for a given fibre loading. Orientation  
 570 polarisation diminishes as frequency increases, and this may  
 571 be attributed to this phenomenon. Orientation polarisation  
 572 takes longer to attain equilibrium than electronic and atomic  
 573 polarisation, and lower frequencies are required for full  
 574 orientation of the molecules. Due to the lag in orientation  
 575 polarisation, as frequency rises, the E decreases.

## 4 Conclusions

576 Phenolic-based hybrid composites were prepared with flax  
 577 fabric and nanofillers. With the addition of nanofillers sig-  
 578 nificant improvements were found. The following conclu-  
 579 sions may be drawn from the data: 580

- Nano-silica was successfully isolated from rice husk. The diffraction peak shows  $2\theta = 22^\circ$  which is specific for nano-silica. Higher thermal stability was observed from TGA. From the FESEM images it shows clusters of nano-silica particles. 581–585
- The XRD data revealed that the incorporation of nano-silica into the PF-flax composite provided efficient reinforcement. 586–587
- The XRD data revealed that the incorporation of nano-silica into the PF-flax composite provided efficient reinforcement. From the mechanical properties, the addition of nano-silica improved the tensile strength, modulus, toughness, and ultimate stress. The improved mechanical characteristics of composites containing 2 NS were attributed to the uniform dispersion of nano-silica in the PF-flax fabric composites. In the presence of 2 NS, an increase in tensile strength of approximately 61% has been recorded. From the fracture surface morphology, it is clear that the absence of visible fibre pull-outs and the debonding of fibres has been significantly reduced by the addition of nano-silica (2 NS). 588–599
- The dielectric constant in PF hybrid composites grows as the loading of nano-silica increases due to polarisation processes 600–601



**Fig. 9** Dielectric properties [a-AC conductivity, b- dielectric constant, c dielectric loss] of PF hybrid composites with nano-silica

602 mediated by nano-silica inclusion. At high frequencies, elec-  
 603 tronic polarisation induces an increase in AC conductivity  
 604 (3 NS). The graph demonstrates that the AC conductivity  
 605 increases as the loading of nano-silica increases. The highest  
 606 dielectric constant values are found in 2 NS.

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 611 work.

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613 **Data Availability** The data that support the findings of this study are  
 614 available from the corresponding author, [Dr. Sreekala M.S.], upon  
 615 reasonable request.

## 616 Declarations

617 **Ethics Approval and Consent to Participate** Not Applicable.

618 **Consent for Publication** Not Applicable.

**Research involving Human Participants and/ or Animals** Not Applicable. 619

**Informed Consent** Not Applicable. 620

**Competing Interests** The authors declare that they have no competing  
 621 interests. 622

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